Dual-layer remote phosphor structure: a novel technique to enhance the color quality scale and luminous flux of WLEDs

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ABSTRACT

The effects of red light-emitting phosphor CaMgSi2O6:Eu²⁺,Mn²⁺ on the optical properties of single-layer remote phosphor structure (SRPS) and dual-layer remote phosphor structure (DRPS) are the focus of this study. The differences in color quality and luminous flux (LF) of white light-emitting diodes (WLEDs) between these two structures are also revealed and demonstrated based on the Mie theory. SRPS consists of one mixed phosphor layer between CaMgSi2O6:Eu²⁺,Mn²⁺ and YAG:Ce³⁺ particles, while DRPS includes two separated layers: red phosphor layer and yellow phosphor layer. In this work, 5% SiO₂ is added into the phosphor layers to increase scattering abilities. Discrepancies in structures greatly affect the optical characteristics of WLEDs. The results showed that the color rendering index (CRI) increased with the concentration in both structures with nearly equal values. Meanwhile, color quality scale (CQS) of DRPS is 74 at ACCTs ranging from 5600K to 8500K, higher than CQS of SRPS which is only 71 at 8500K. In addition, the luminous flux of DRPS is significantly higher than SRPS at 2% -14% of CaMgSi2O6:Eu²⁺,Mn²⁺. In summary, DRPS is better for color quality and lumen output in comparison to SRPS and adding the right amount of red phosphor can enhance CQS and LF.

Keywords:
- Color rendering index
- Luminous flux
- Dual-layer phosphor
- Mie-scattering theory
- WLEDs

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1. INTRODUCTION

The light-emitting diodes that eject white light (WLEDs) are one of the most commonly used lighting solution in the lighting and display systems industry for their unique properties, including reliability, luminous efficiency, and energy efficiency, which is likely to replace older lighting techniques in the future [1-4]. However, there are major changes in WLEDs that can be made to reach the full potential including phosphor conversion and luminous efficiency [5-6], as well as improving color purity and price competitiveness against incandescent and fluorescent lamps [7-10]. In previous studies, quantum efficiency was enhanced using a phosphors synthesis methodology called exploiting flux [11-14] and inventing a novel phosphor. Researchers have recently adjusted the configuration of LED tools by modifying the gap between the phosphor material and the lighting emitting chip, creating a new type of LED named remote-type LED.
which works on the scattered photon extraction concept and improves the quantum efficiency of WLED up to
60% by boosting the back-scattered photons extraction efficiency [15].

Enhancing brightness and luminous efficiency also relies upon the color quality. A notable method that
usually applied in WLED production is emerging the blue light of InGaN LED chip with the yellow
green light from a phosphor; however, the red light lack these elements [16]. Therefore, it is essential that red
phosphor and green-yellow phosphor are combined in order to yield better color quality. The process that
produce the compound of these two phosphors, however, damages the obtained light from emission process
due to a part of light emitted from one phosphor is reabsorbed by another causing an overlap of phosphor
spectrawhich leads to low color quality [17]. A possible solution is stacking the layers of phosphor materials
in the order of red phosphor layer below the green one to enhance the efficiency. This pattern suggests that
the layer of red phosphor can reduce the reabsorption of light from green phosphor if placed directly above
the chip [18]. Nevertheless, stacked phosphor layers model cannot be re-implemented for copyright and
patents reasons to avoid risks related to law violations, [8], which forces us to look for a new method named
phosphor-in-glass (PiG) to improve the chromatic quality and heat performance in WLED configuration with
remote phosphor layers [19-20]. Not only does PiG benefit the luminous efficacy and thermal stability of
WLEDs but it also offers color quality advantages that very few research papers have mentioned or proved.
In this study, optical properties such as luminous efficiency and color quality of WLEDs are improved by
utilizing a new technique called dual-layer remote phosphor structure (DRPS). Besides, the comparison of
optical characteristics between SRPS (single remote phosphor structure) and DRPS is also clearly presented
and demonstrated based on Mie scattering theory.

2. DETAIL OF SIMULATION

2.1. Simulation of SRPS and DRPS

As can be seen from Figure 1(a), the WLED simulation with remote phosphor structures that have
the average CCTs of 5600 K, 6600 K, 7000 K, 7700 K, and 8500 K was made from a commercial software
utilizing the Monte Carlo ray-tracing method, the LightTools 8.1.0. Figure 1(b) depicted the 3-D simulation
physical model of WLEDs used to perform optical simulations of remote package WLEDs. SRPS consists of
a single phosphor layer in which CaMgSi2O6:Eu2+,Mn2+ are mixed with YAG:Ce3+ compound as illustrated
in Figure 1 (c). On the other hand, according to Figure 1 (d), DRPS includes two separated layers of red
phosphor and yellow phosphor. This realistic model of WLEDs consists of an 8 mm long reflector at the sole,
2.07 mm high, and 9.85 mm long at its top surface. As depicted in Figure 1(d), the thickness for each layer
of remote phosphor is 0.08 mm with nine LED chips, each of them has 1.16 W radiant power at 455 nm
wavelength, a square sole of 1.14 mm, and is 0.15 mm high bounded to the spaces at the bottom of
the reflector. When the SiO2 concentration in the mixture of phosphor is fixed at 5%, the concentration of
CaMg phosphor climbs ceaselessly from 2% to 30%. However, the average CCT values are stabilized with
the modification of YAG:Ce3+ wt.

![Figure 1. a) Photograph of WLEDs sample, (b) Simulation of the WLEDs using LightTools program,
(c) Illustration of SRPS with SiO2 (green) and CaMgSi2O6:Eu2+,Mn2+ (red) in YAG:Ce3+ compound (yellow),
(d) Illustration of DRPS with SiO2 in CaMgSi2O6:Eu2+,Mn2+ compound and YAG:Ce3+ compound.](image)

Next, the LightTools 8.5.0 software is applied to simulate the particles properties of SiO2 and
CaMgSi2O6:Eu2+,Mn2+. The refractive index of SiO2 and CaMgSi2O6:Eu2+,Mn2+ phosphor is 1.54 and 1.80
respectively. The SiO2 particles have spherical shape with a 3 μm average radius in the Mie simulation while
the YAG:Ce3+ phosphor particles have an average radius of 7.25 μm with consistent 1.38 refractive index at
every wavelengths, and the refractive index for the silicone glue is 1.5. To keep the average CCT value steady,
it is vital to vary the diffusional particle density in the direction of increasing the weight of the diffuses to
increase their percentage while lessening the weight of YAG:Ce3+ phosphor to retain the balance.
2.2. Preparation of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ phosphor

Ordinarily, red-emitting phosphor CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ is suitable for high loading and long lifetime fluorescent lamps because of the outstanding properties in quantum efficiency or chemical and thermal stability [11]. The CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ composition, which is supposedly a substance that can enhance the luminous efficacy of WLEDs, is created through the chemical processes between different materials such as CaO, MgO, SiO$_2$, Eu$_2$O$_3$, MnCO$_3$, and NH$_4$Cl. The CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ phosphor composition is calculated in detail as can be seen from Table 1. The fabrication of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ must be taken in order of six separate steps: mixing, drying, firing twice, washing and drying again to get the best result, each of them plays an essential role and closely relates to the preceding step(s). This order is presented specifically as follows: (1) Mix the materials by slurrying them in methanol with the focus on soaking with a few cubic centimeters water; (2) Dry the mixture in the air until it completely gets drained; (3) Fire the dried materials in capped quartz tubes under the condition of N$_2$ at 1000ºC within an hour; (4) Fire this compound again in capped quartz tubes n an hour but with CO at 1150ºC. (5) Collect and wash with water to remove residue and dust. (6) Wait until the materials dry and we achieve CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$.

<table>
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<th>Ingredient</th>
<th>Mole (%)</th>
<th>By weight (g)</th>
<th>Molar mass (g/mol)</th>
<th>Mole (mol)</th>
<th>Ions Mole (mol)</th>
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<td>--</td>
<td>Mn$^{2+}$</td>
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3. SCATTERING COMPUTATION

The scattering coefficient $\mu_{sca}(\lambda)$, anisotropy factor $g(\lambda)$, and reduced scattering coefficient $\delta_{sca}(\lambda)$ according to Mie theory [21-25] can be defined by applying expressions (1-3):

$$\mu_{sca}(\lambda) = \int N(r)C_{sca}(\lambda, r)dr$$

(1)

$$g(\lambda) = 2\pi \int_{-1}^{1} p(\theta, \lambda, r) f(r) \cos \theta d \cos \theta dr$$

(2)

$$\delta_{sca} = \mu_{sca} (1 - g)$$

(3)

where $N(r)$ indicates the density of distributed diffusional particles (mm$^{-3}$), $C_{sca}$ is the scattering cross sections (mm$^2$), $p(0, \lambda, r)$ is the phase function, $\lambda$ is the light wavelength (nm), $r$ is diffusional particles radius (µm), $\theta$ is the scattering angle (°C), and $f(r)$ is the function describe the diffusor distribution of size in the phosphor layer, which can be calculated as follows:

$$f(r) = f_{dif}(r) + f_{phos}(r)$$

(4)

$$N(r) = N_{dif}(r) + N_{phos}(r)$$

$$= K_N \{ f_{dif}(r) + f_{phos}(r) \}$$

(5)

The function $N(r)$ are compiled by the density of diffusive particle $N_{dif}(r)$ and phosphor particle $N_{phos}(r)$. $f_{dif}(r)$ and $f_{phos}(r)$ are functions for diffusor and phosphor particle size distribution. $K_N$ is the diffusor unit in one diffusor concentration and is defined by:

$$c = K_N \int M(r) dr$$

(6)

where $M(r)$ is the diffusive unit spatial distribution of mass, proposed by equation:
\[ M(r) = \frac{4}{3} \pi r^3 [\rho_{\text{diff}} f_{\text{diff}}(r) + \rho_{\text{phos}} f_{\text{phos}}(r)] \]

\( \rho_{\text{diff}}(r) \) and \( \rho_{\text{phos}}(r) \) describe the diffusor and phosphor density.

In Mie-scattering theory, \( C_{\text{ sca}} \) can be achieved as a results of the equation below:

\[ C_{\text{ sca}} = \frac{2\pi}{k^2} \sum_{n=0}^{\infty} (2n-1)\left|a_n^2 + b_n^2\right| \]

where \( k = \frac{2\pi}{\lambda} \), and \( a_n \) and \( b_n \) are estimated by:

\[ a_n(x, m) = \frac{\psi_n^{(m)}(x)\psi_n(x) - m\psi_n^{(m)}(x)\psi_n(x)}{\psi_n^{(m)}(x)\psi_n(x) - m\psi_n^{(m)}(x)\psi_n(x)} \]

\[ b_n(x, m) = \frac{m\psi_n^{(m)}(x)\psi_n(x) - \psi_n^{(m)}(x)\psi_n(x)}{m\psi_n^{(m)}(x)\psi_n(x) - \psi_n^{(m)}(x)\psi_n(x)} \]

Which have the refractive index \( x = k \cdot r \cdot m \) and Riccati - Bessel function \( \psi_n(x) \) and \( \xi_n(x) \). It can be seen from Figure 2 that the scattering coefficients at 453 nm, 555 nm, and 680 nm wavelengths increase dependently on the \( \text{CaMgSi}_2\text{O}_6:\text{Eu}^{2+},\text{Mn}^{2+} \) phosphor concentration. The scattering effects of \( \text{CaMgSi}_2\text{O}_6:\text{Eu}^{2+},\text{Mn}^{2+} \) and \( \text{SiO}_2 \) particles cause considerable influence on RP-WLEDs. Particularly, the red light from \( \text{CaMgSi}_2\text{O}_6:\text{Eu}^{2+},\text{Mn}^{2+} \) has better absorbing capability than the light radiation of LED and therefore, the will be abundant red light which can be used to compensate for deficiency in RP-WLEDs. Furthermore, 5% wt. of \( \text{SiO}_2 \) was added to enhance the scattered light, resulting in a rise assimilated light of pc-LEDs, which is the reason why \( \text{CaMgSi}_2\text{O}_6:\text{Eu}^{2+},\text{Mn}^{2+} \) and \( \text{SiO}_2 \) particles are utilized to yield white LEDs with better color quality. Next, the anisotropy factor of \( \text{CaMgSi}_2\text{O}_6:\text{Eu}^{2+},\text{Mn}^{2+} \) particles for wavelengths of 453 nm, 555 nm, and 680 nm was presented in Figure 3, which indicates the anisotropy factor values at 680 nm wavelength increases in comparison to 555 nm values. In comparison with other wavelengths, the anisotropy factor values obtained at 453 nm wavelength is the highest, which means \( \text{CaMgSi}_2\text{O}_6:\text{Eu}^{2+},\text{Mn}^{2+} \) particles are favorable to the color homogeneity of remote phosphor WLEDs.

![Figure 2. Scattering coefficients of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ of 453 nm, 555 nm, and 680 nm wavelengths](image)

![Figure 3. Anisotropy scattering of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ of 453 nm, 555 nm, and 680 nm wavelengths](image)

The refractive index of silicone (\( n_{\text{sil}} \)) in SRPS and DRPS is 1.53 while \( n_{\text{phos}} \) is the refractive index of the phosphor particles. Therefore, the relative refractive values of diffusive particles (\( n_{\text{sil}} \)) and phosphor (\( n_{\text{phos}} \)) in the silicone can be computed from \( m_{\text{diff}} = n_{\text{diff}} / n_{\text{sil}} \) and \( m_{\text{phos}} = n_{\text{phos}} / n_{\text{sil}} \). Then the phase function is expressed as follows:

where $\beta(\theta, \lambda, r)$, $S_1(\theta)$ and $S_2(\theta)$ are the angular scattering amplitudes calculated by the following equations:

$$\beta(\theta, \lambda, r) = \frac{1}{2} \left( |S_1(\theta)|^2 + |S_2(\theta)|^2 \right)$$  \hspace{1cm} (12)

$$S_1 = \sum_{n=1}^{\infty} \frac{2n+1}{n(n+1)} \left[ a_n(x, m) \pi_n(\cos \theta) + b_n(x, m) \tau_n(\cos \theta) \right]$$  \hspace{1cm} (13)

$$S_2 = \sum_{n=1}^{\infty} \frac{2n+1}{n(n+1)} \left[ a_n(x, m) \pi_n(\cos \theta) + b_n(x, m) \tau_n(\cos \theta) \right]$$  \hspace{1cm} (14)

### 4. RESULTS AND DISCUSSION

Figure 4 showed that the reduction in scattering coefficient of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ at 453nm, 555nm, and 680nm wavelengths are nearly equal, creating the scattering stability of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$, which is beneficial for the chromatic performance of remote phosphor WLEDs. The angular scattering amplitudes of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ are also computed by MATLAB program after that. According to the obtained results, CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ particles are greatly advantageous to the blue-light scattering. As known, a large amount in blue light can lessen the effect of yellow ring. Meanwhile, CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ particles not only compensate to the red-light but also to the blue-light, as shown in angular scattering amplitudes within Figure 5. These calculations demonstrating the results from Figure 6 to Figure 8.

![Figure 4. Reduced scattering coefficient of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ of 453nm, 555nm, and 680nm wavelengths](image)

![Figure 5. The angular scattering amplitudes of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ of 453nm, 555nm, and 680nm wavelengths](image)

The discrepancy in CRI values between the SRPS and DRPS shown in Figure 6 is negligible. When increasing the concentration of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ up to 22%, CRI of SRPS and DRPS tend to climb up, especially with the highest ACCT at 8500 K, which have an important meaning in improving CRI for both SRPS and DRPS. While CRI is hard to control at high ACCT (8500 K), phosphor CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ can definitely make it possible. However, CRI is just one of the indicators used to evaluate the color quality, which has recently been replaced by a newer quality indicator called CQS.
Figure 6. CRI values of SRPS (a) and DRPS (b) grow with CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ concentration

CQS involves three elements: CRI, viewers' preference, and color coordinates, making it become the most critical indicator to assess color quality. As known, the CRI value of SRPS continues to increase when CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ concentration exceeds 30% while the CRI value of DRPS has a tendency to drop at all ACCTs after this concentration limit. This is the same to CQS as the results of SRPS and DRPS CQS interactions with CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ concentration shown in Figure 7. SRPS can yield the highest CQS of 71 at CCT 8500K while 74 is the highest CQS value of DRPS at most ACCTs. Thus, it can be concluded that DRPS brings greater color quality than SRPS. As illustrated in Figure 6 and Figure 7, CRI and CQS go up sharply when the red phosphor concentration is adjusted in the range of 2% - 30%. The greatest value of CRI and CQS were respectively 85 and 71 at 30% CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn concentration while the highest angular scattering amplitudes of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn is achieved at 453 nm wavelength as mentioned in Figure 5, which means the blue-light scattering is benefited from the added CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn concentration as well, bringing not only advantages for chromatic performance improvements but also light output. The backscattering event, however, will occur in SFPS if CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn concentration exceeds 14% causing light loss and resulting in reduced luminous flux which can be observed from the lumen output values of SRPS and DRPS expressed in Figure 8.

Figure 7. CQS values of SRPS (a) and DRPS (b) grow with CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ concentration

As a result, the luminous flux increases sharply at first, then reaches the peak value and finally goes down slightly, as depicted in Figure 8 (a). Figure 8 (b) depicted a sharp reduction in DRPS brightness when increasing the concentration of red phosphor CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$, which is the result of a noticeable decrease in energy from transmitting light through the red phosphor layer. However, at 2-14% CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$, luminous flux of DRPS is always greater than SRPS at all ACCTs. Therefore, DRPS provides both higher brightness and CQS for WLEDs than SRPS.
5. CONCLUSION

To sum up, this study has focused on solving two problems: (1) comparing CQS and LED optical performance of SRPS and DRPS, and (2) analyzing the effect of red phosphor CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ on CQS and LF of these two structures. Both phosphor structure and phosphor concentration should be simultaneously modified until an appropriate value to achieve the desired CQS and LF. The results showed that CRI and CQS go up when raising the CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ concentration. DRPS yields a CQS of 74 at ACCTs ranging from 5600K to 8500K but it causes a significant decrease in LF. However, this structure always brings higher luminous flux than SRPS does at all ACCTs with about 2-14% of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ concentration. DRPS yields a CQS of 74 at ACCTs ranging from 5600K to 8500K but it causes a significant decrease in LF. However, this structure always brings higher luminous flux than SRPS does at all ACCTs with about 2-14% of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ concentration. This result is demonstrated through the scattering properties of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ including the scattering coefficient $\mu_s(\lambda)$, anisotropy factor $g(\lambda)$, reduced scattering coefficient $\delta(\lambda)$, and the angular scattering amplitudes $S_1(\theta)$ and $S_2(\theta)$. In conclusion, the CRI of SRPS and DRPS are almost equivalent while the CQS and LF of DRPS outperform that of the SRPS. Therefore, the concentration of CaMgSi$_2$O$_6$:Eu$^{2+}$,Mn$^{2+}$ must be selected appropriately to achieve the desired CQS and LF.

REFERENCES


